

Raising Medium Field Q_0 : Ways to Proceed

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Outline

- Importance of Q_o for CW applications
- State of the art in SRF niobium cavities
- Contributors to medium field Q_o
 - Proposed models and recent developments
- Possible ways to increase Q_o
- Proposed research effort

Project X (CW): Q_0 is crucial

$$Q_0 = \omega U / P_{\text{diss}} = G / R_s$$



[Equality only holds for $R_s(H) = \text{const}$]

Surface resistance

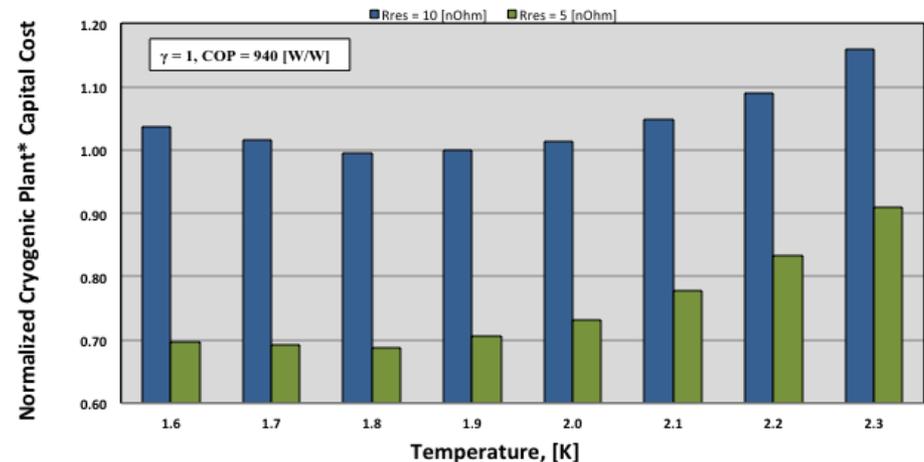
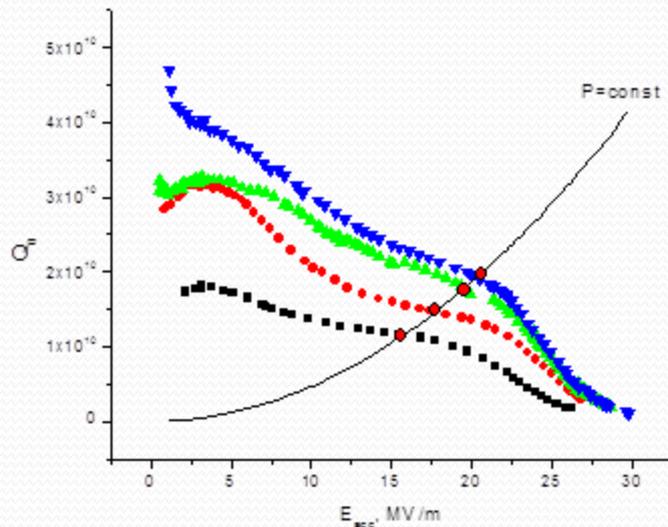
Q -slopes



$$R_s = (R_{\text{BCS}}(T) + R_{\text{res}}) * f(H)$$

Power dissipated in cavity walls $P_{\text{diss}} \Rightarrow$ determines dynamic heat load \Rightarrow requirements for cryogenic power plant

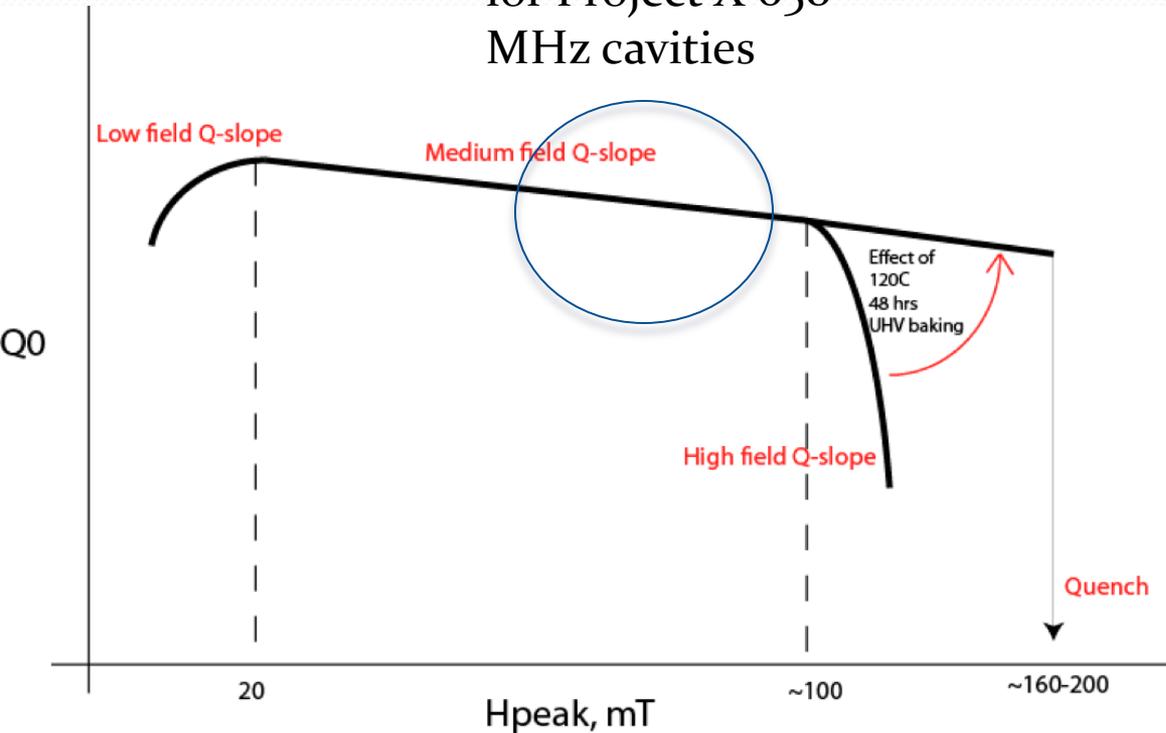
Project X cryoplant cost – determined by Q_0
[A. Klebaner et al, PX 2011 Collab. Meeting]



* - plant cost is approximately 75 % of the total cryogenic system cost

Excitation curve $Q(H_{\text{peak}})$

Operational gradients
for Project X 650
MHz cavities



- Three distinct regions of the Q-curve – low, medium, and high field Q-slope
- Respond differently to heat and chemical treatments
 - Most likely different origin
- ILC standard processing – EP+120C to overcome high field Q-slope to reach **highest gradient**
- **THERE IS NO OPTIMIZED RECIPE FOR MEDIUM FIELD Q YET**
 - Interest has grown only recently, nobody really worked on it

Factors affecting Q_0

- Many phenomena understood and controlled
 - Multipacting
 - Field emission
 - Hydrogen Q-disease
- Main problem – lack of clear **understanding** and control of remaining ones
 - Residual resistance
 - Low, medium, high field Q-slopes

Residual resistance

A few mechanisms have been identified:

- Trapped flux due to residual magnetic field on cooling
 - Minimized by shielding
 - Is there more we can do? Maybe modifying things from the material point of view? Like decreasing pinning or maybe increasing depending on what the dissipation mechanism is
- Niobium hydrides
 - More on this later
- Condensed gases
- All contributing factors not yet uncovered – no control based on knowledge

Medium field Q-slope (MFQS)

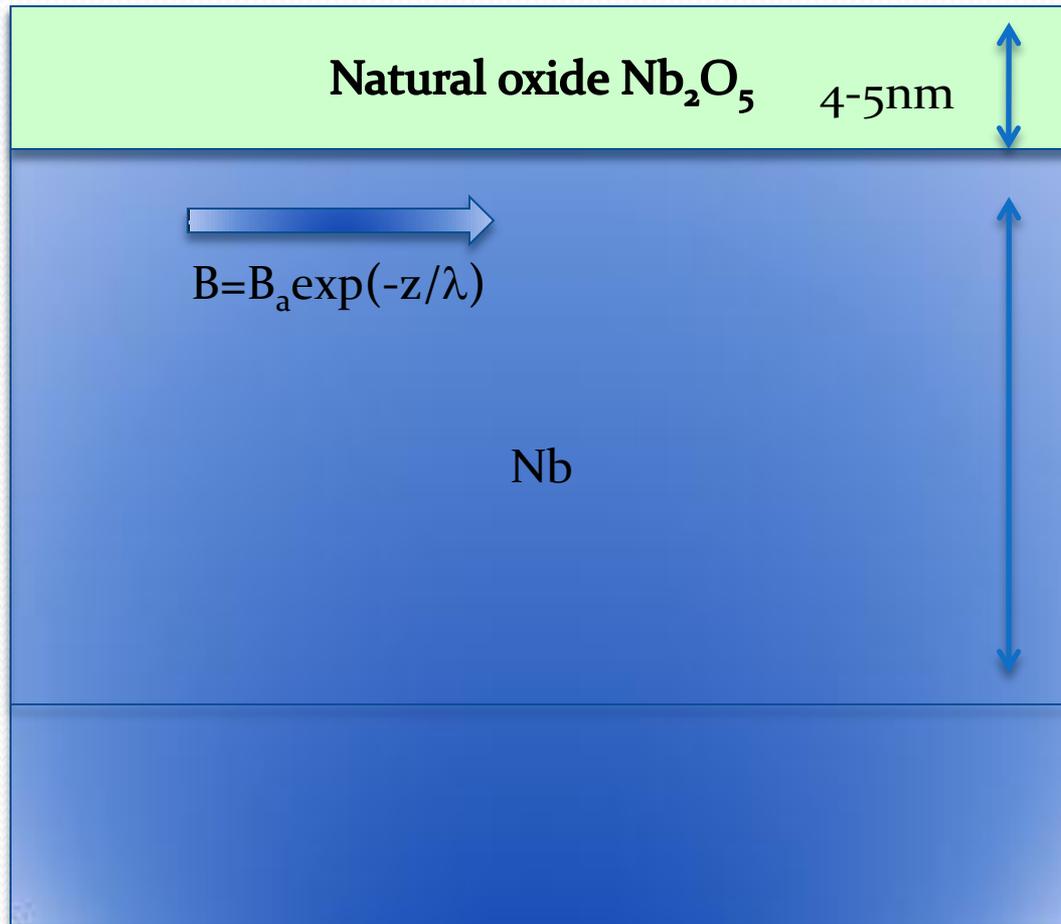
- Historically proposed mechanisms include
 - Thermal feedback
 - $R_{BCS} \sim \exp(-1/T) \Rightarrow T \uparrow \Rightarrow R_{BCS} \uparrow$ – positive feedback
 - Problem – underestimates the slope, not the only player
 - Hysteretic losses due to Josephson fluxons penetrating at “weak links” (came from the observation that often MFQS has linear component in it)
 - $R_s = R_{s0}(1 + b * H/H_c)$
 - Non-linear BCS
 - $\Delta(v_s) = \Delta - p_f |v_s| \Rightarrow$ decreased gap \Rightarrow
 $R_s = R_{s0}(1 + C(\Delta/T)^2(H_o/H_c)^2)$
 - Problem – overestimates the slope

Newer findings

- Hydrides might actually ALWAYS be present – strong effect on medium field Q via R_{res} , may also be $R_{\text{res}} = R_{\text{res}}(\text{H})$
 - A lot of hydrogen in all samples near surface (ERDA studies, Romanenko, Goncharova, SRF'2011, submitted to PRSTAB)
 - Hydride precipitates identified and directly observed for the first time in a single cell (Cutout studies, Romanenko et al, SRF'2011)

Cavity surface

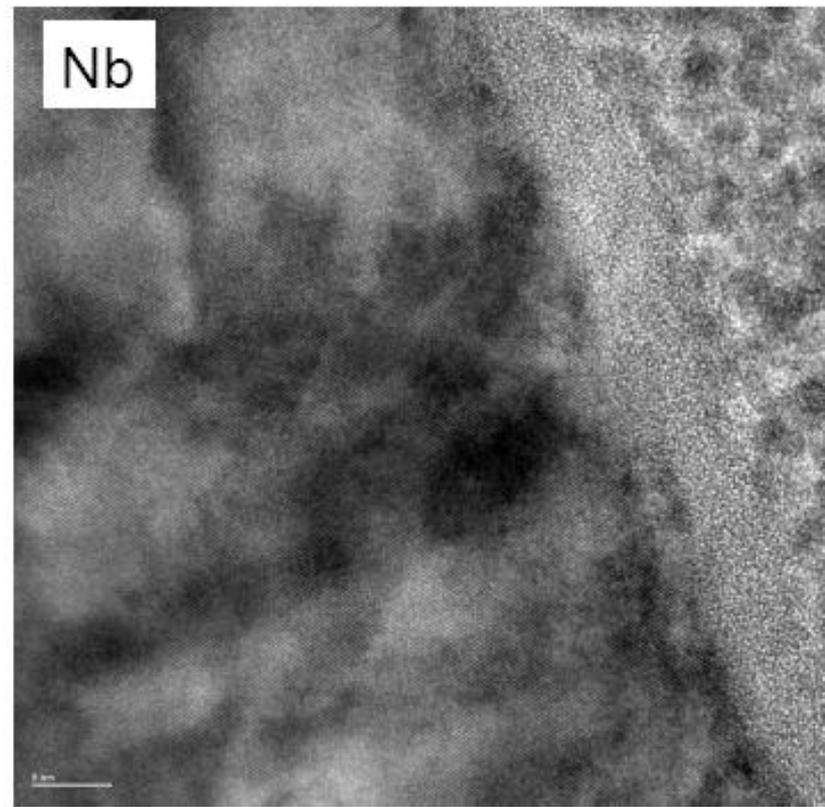
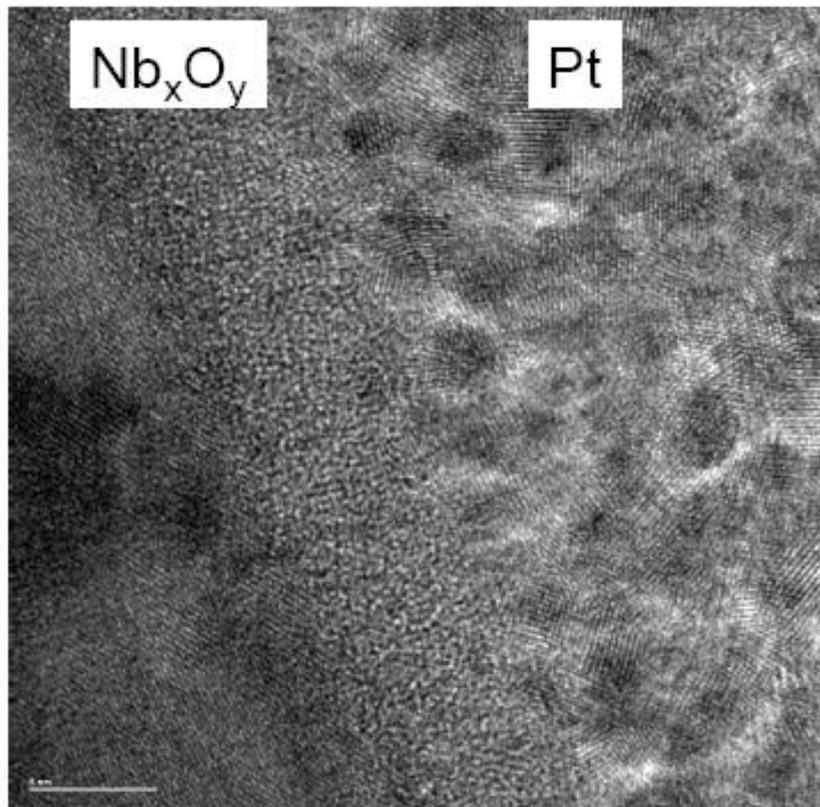
Z



Processes, which can modify this layer in a controlled manner are of special interest

Compare: "Light EP" or 20 um removal = approximately 500 penetration depths - no control

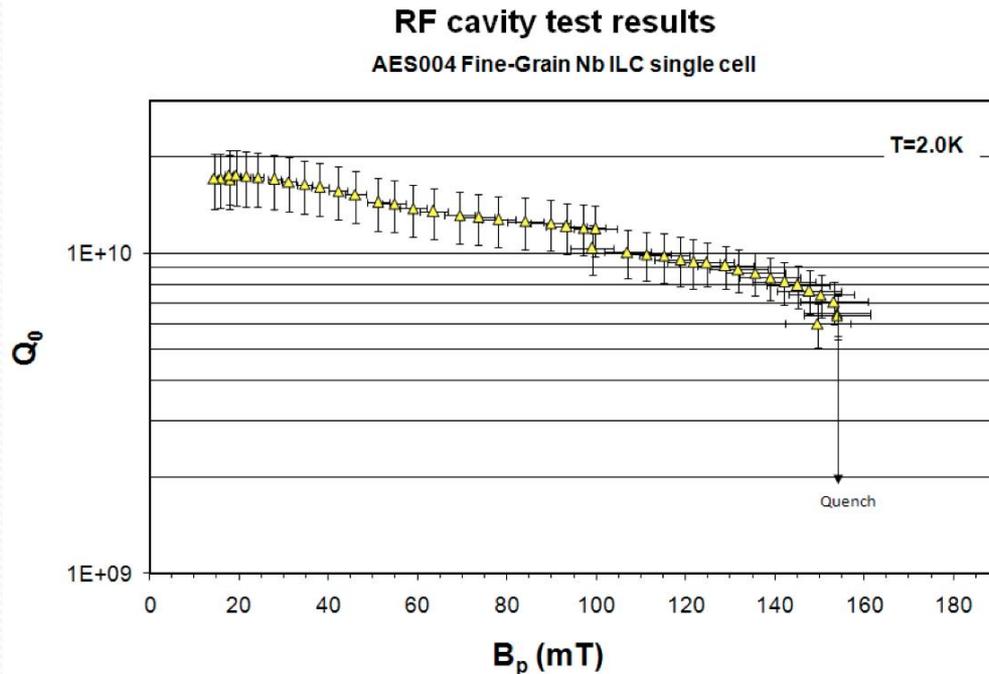
TEM images of Nb near-surface



[Romanenko et al, SRF'2009, TuOAAU02]

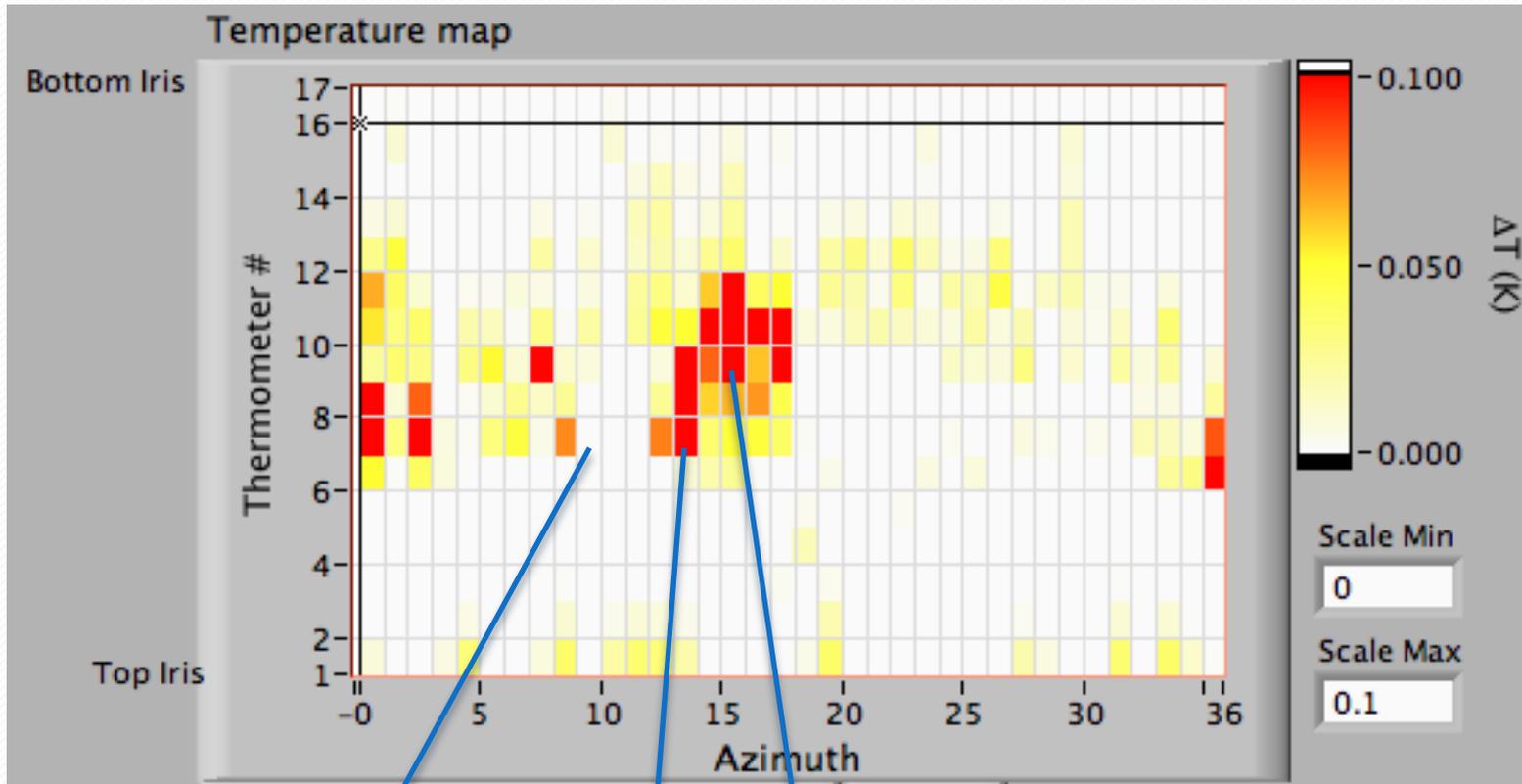
Unexpected finding - HYDRIDES

- 1.3 GHz fine grain single cell
 - EP+120C bake at ANL/FNAL
 - RF tested at JLab with thermometry last year (collaboration with G. Ciovati)



Thermometry system attached to outside cavity walls

Lossy areas AFTER mild baking

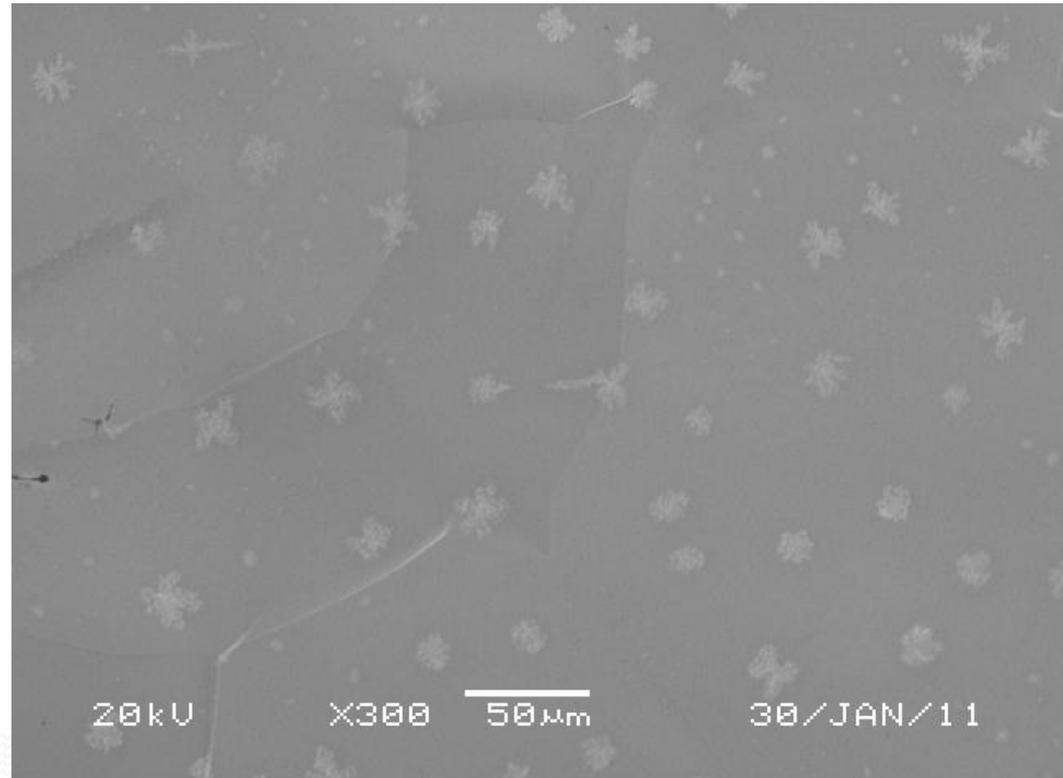
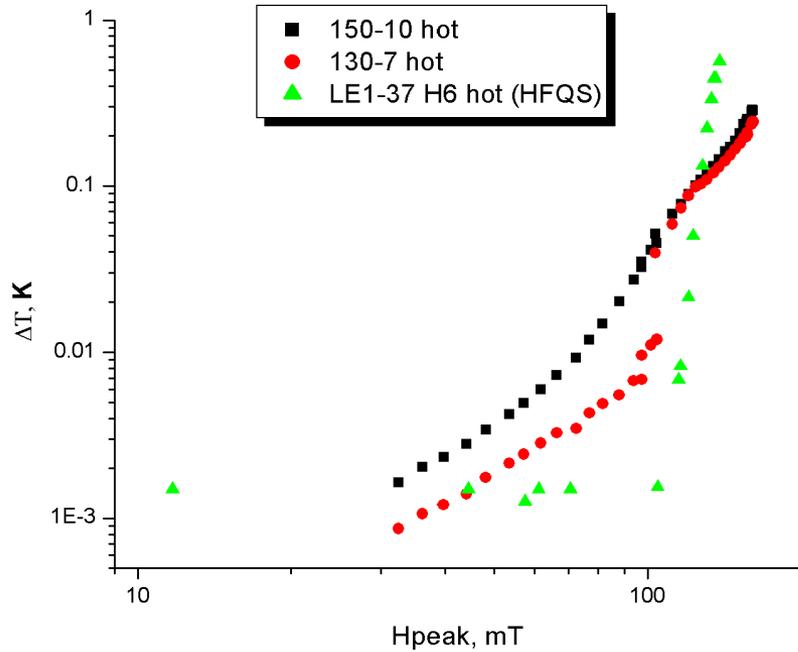


90-7 cold spot

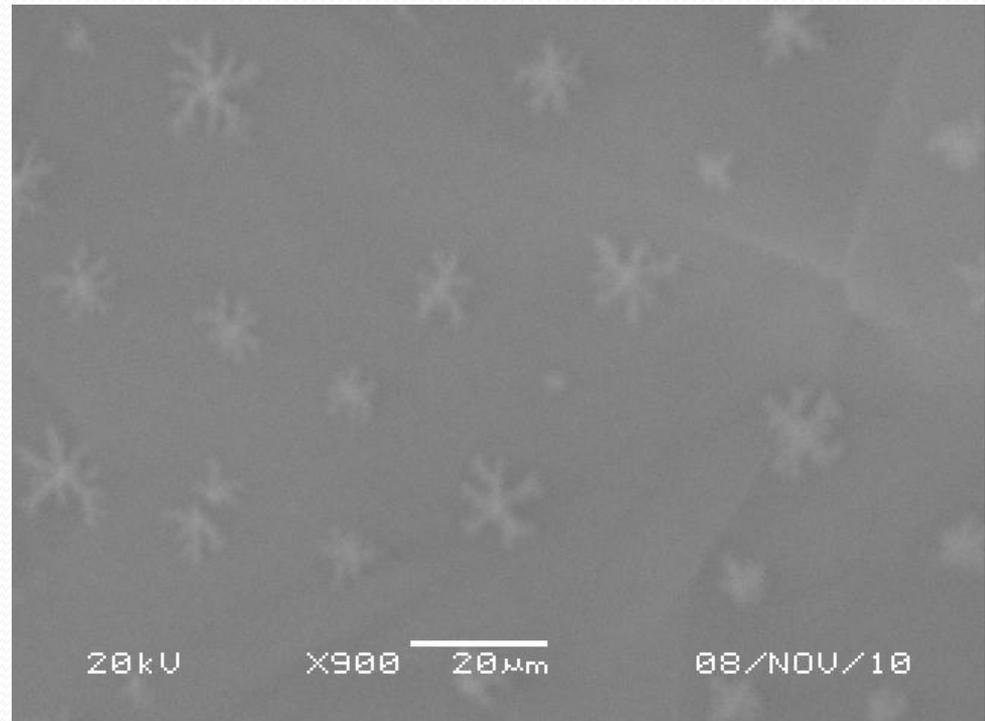
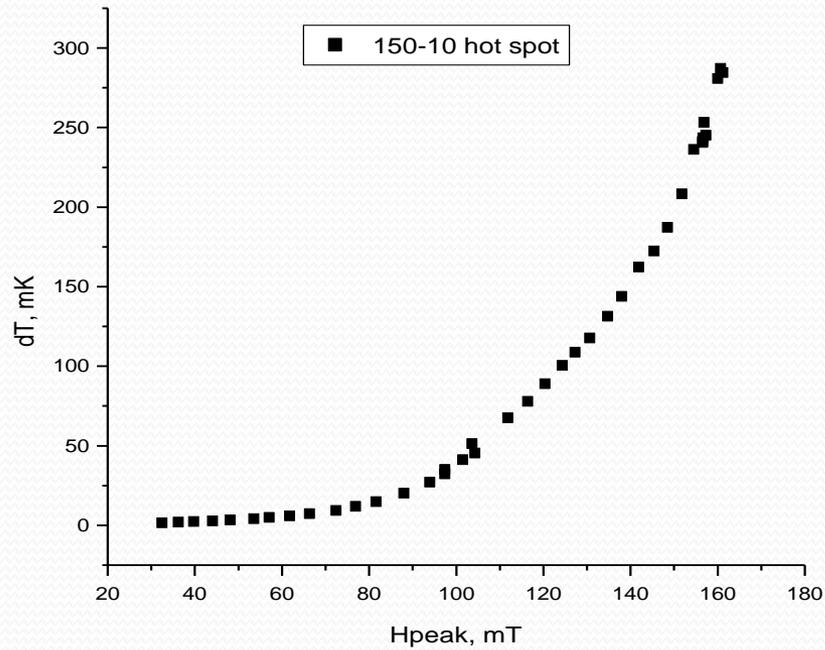
130-7 hot spot

150-10 hot spot

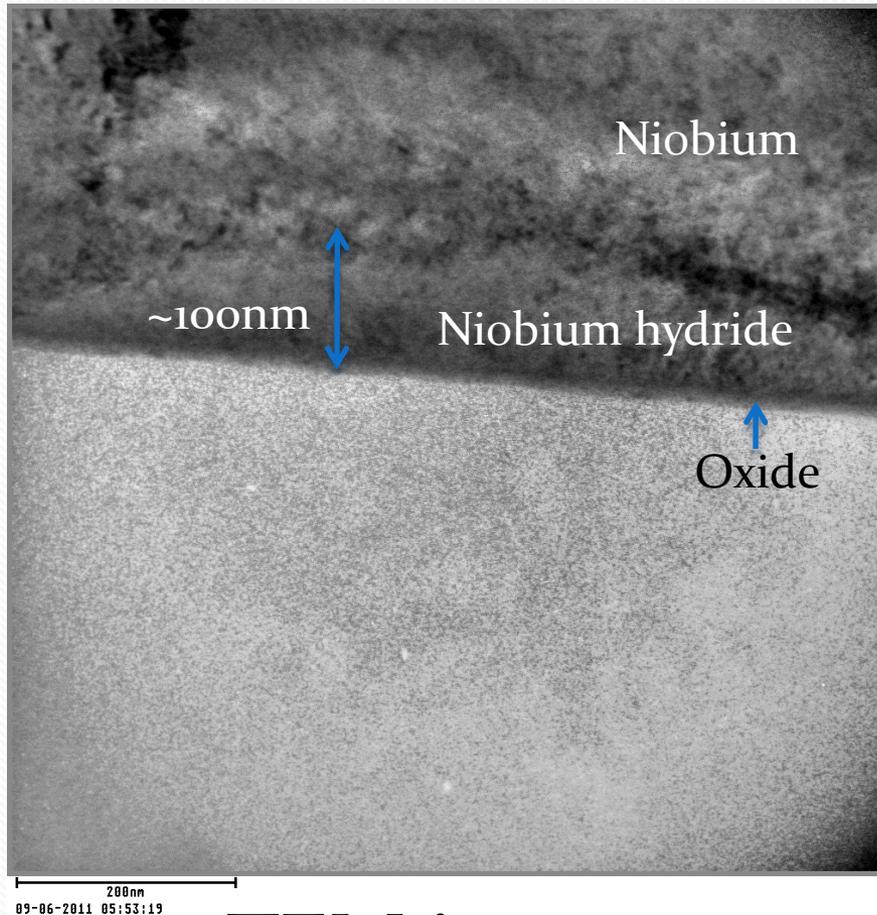
130-7 hot spot



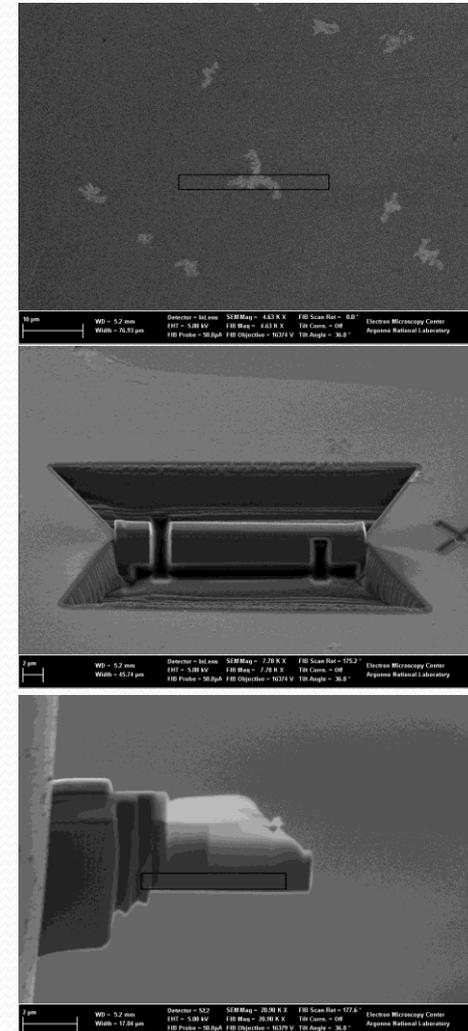
150-10 hot spot



TEM: detailed structure of “stars”

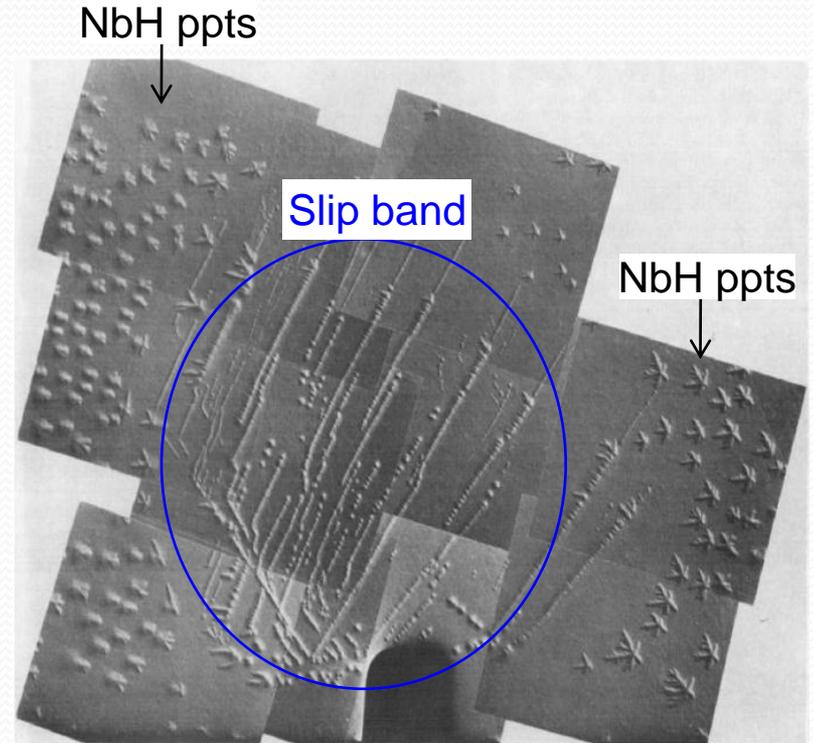
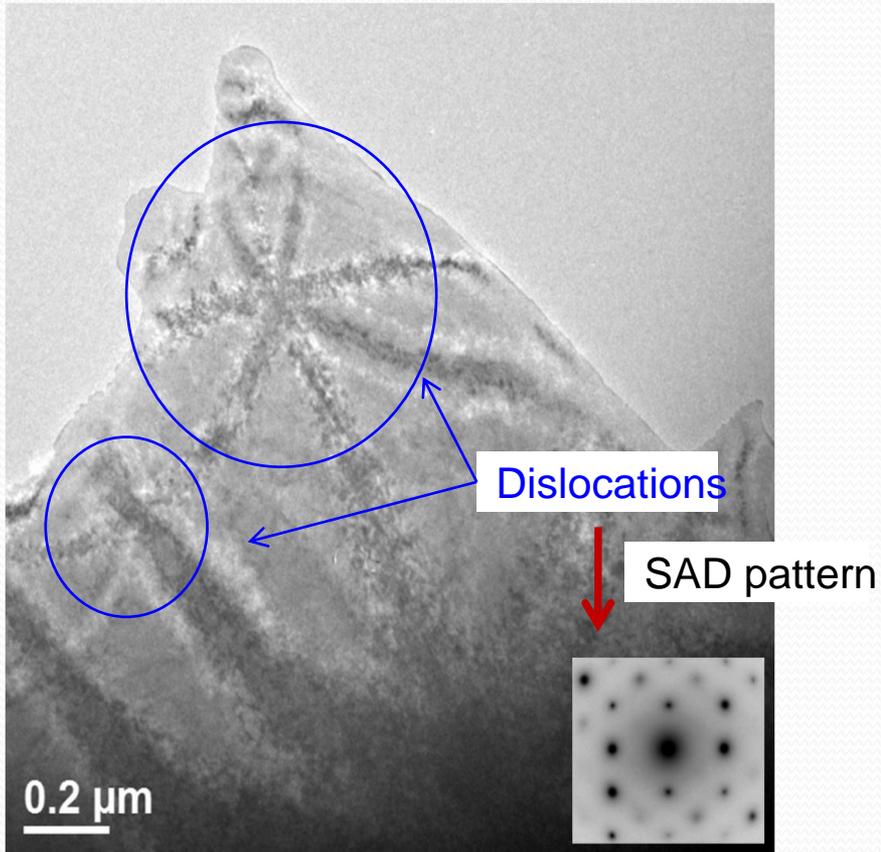


TEM image



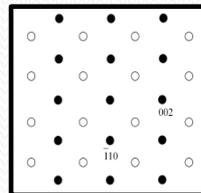
Making
TEM
sample
from
“star” by
FIB

◆ NbH precipitates near dendritically shaped dislocations



Grossbeck, Birnbaum, Acta Metall. 25 (1977) pp. 135

Nb 110 zone
Simulated SAD
pattern

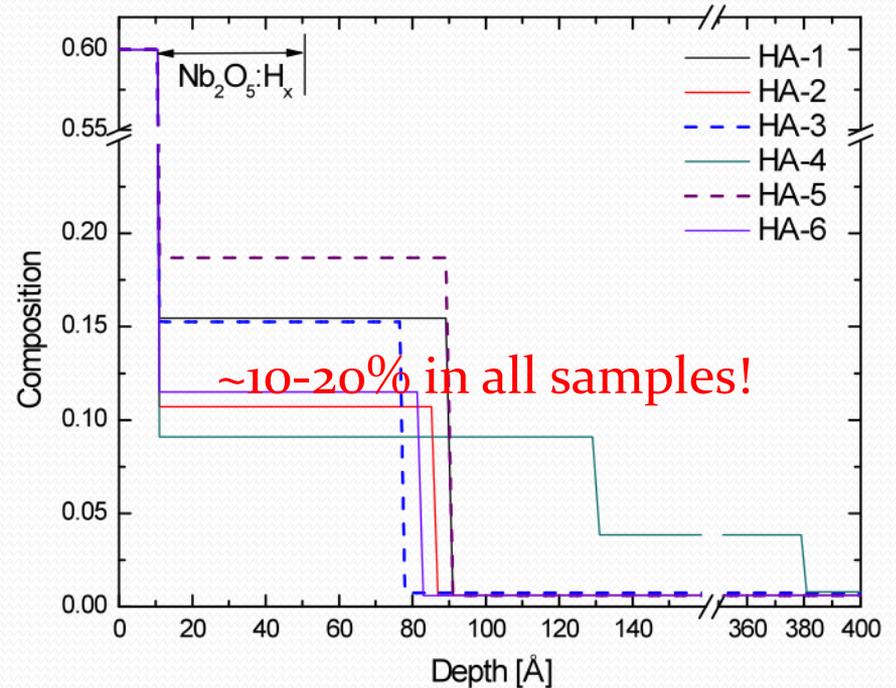
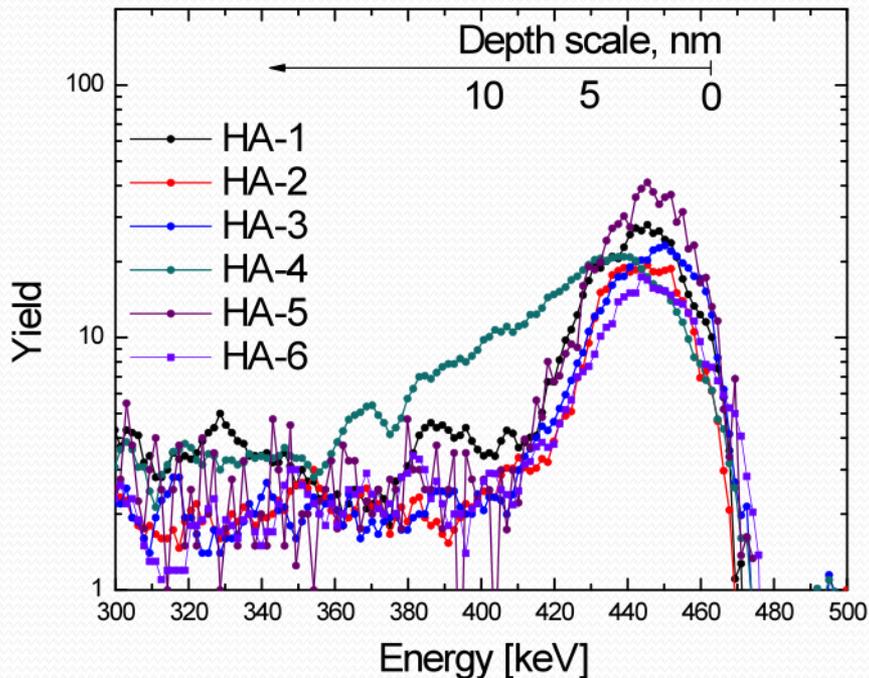
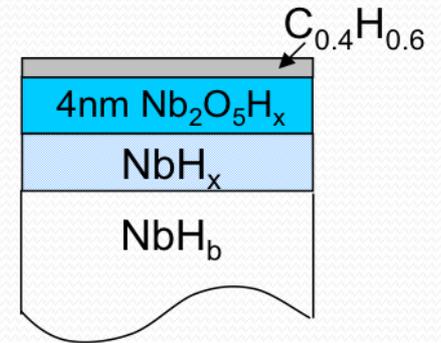


● Nb ○ H sublattice

Hydrogen near-surface enrichment

- HA-1->BCP
- HA-2->+800C
- HA-3->+120C
- HA-4->+HF
- HA-5->BCP+600C
- HA-6->+120C

Elastic recoil detection shows hydrogen enrichment in all samples



Hydrides – major player in R_{res} and MFQS?

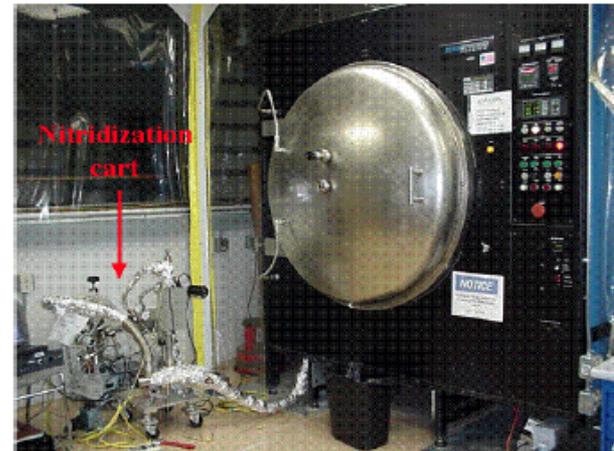
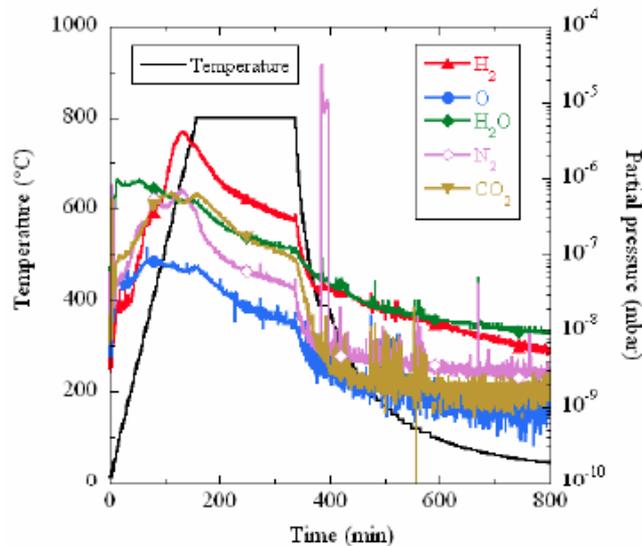
- A lot of hydrogen in the near surface
 - Must precipitate on cooldown
 - Form of precipitation depends on the process and presence of nucleation centers
- Beta phase of NbH_x normal conducting $T > 1.3K$
 - May become superconducting by proximity effect depending on the size of precipitates at higher T if surrounded by SC Nb

Q₀ R&D – other labs

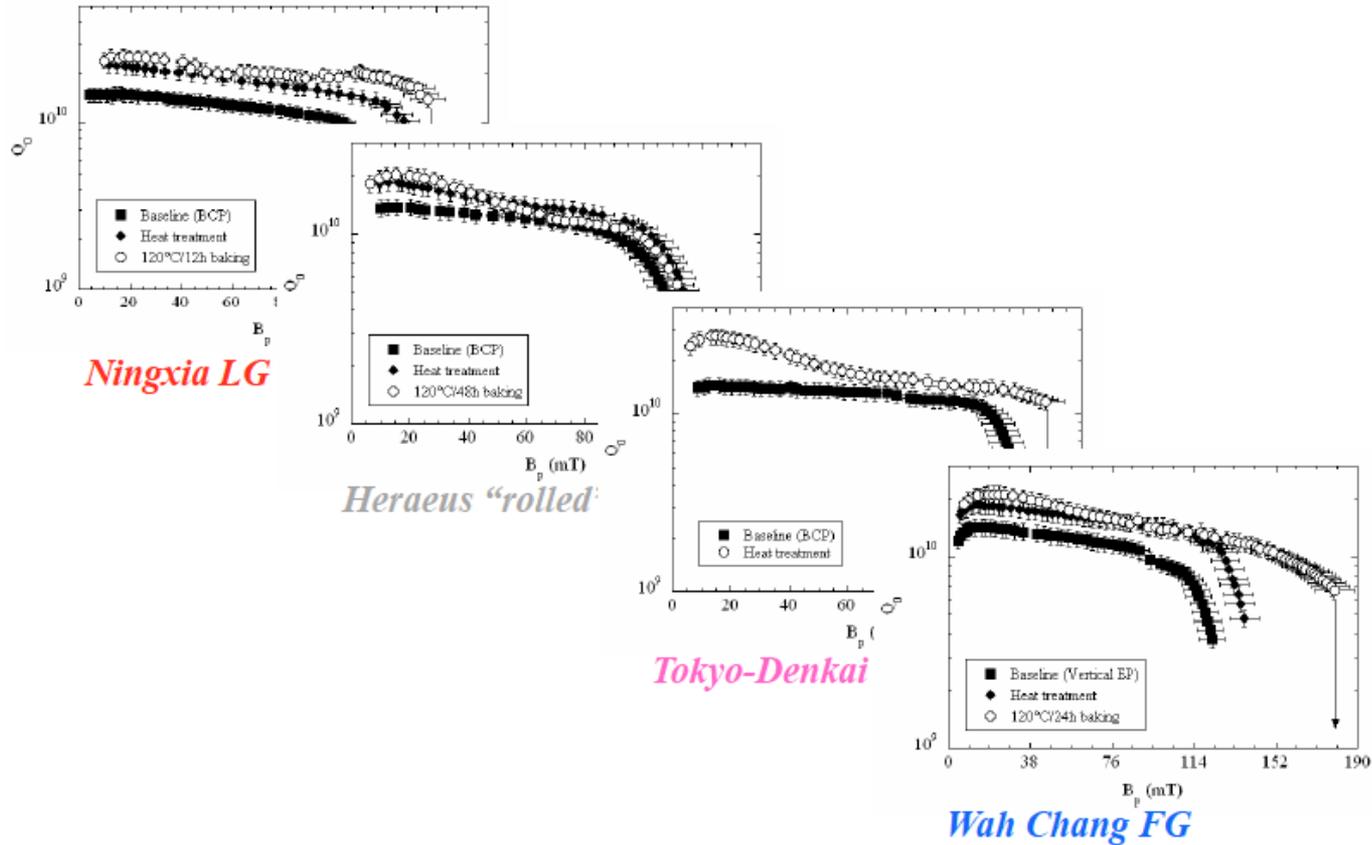
- Only JLab has related R&D plans (to the extent of my knowledge)
- Jlab Plan: exploration of heat treatment/nitride passivation with the dedicated furnace
 - Anneal at 800+C to remove interstitial impurities and lattice defects in the near-surface layer
 - Prevent defects/interstitials from reforming on cooldown by “capping” with niobium nitride at intermediate temperature (400C)
- Preliminary results reported at SRF Materials Workshop 2010, Tallahassee by G. Ciovati
- Spiral 2 has huge improvement in MFQS at 120C baking, this seems to be true for many low beta cavities – importance of low T heat treatments systematic studies in looking for the best recipe

Heat treatment procedure

- UHV Heat treatment at 800 °C/3h
- Rapid cooling to 400 °C, admit $\sim 5 \times 10^{-6}$ Torr N₂ for 15 min
- Cool to 120 °C and hold for 12 h (optional)
- No chemical etching afterwards!!! Just degreasing and HPR



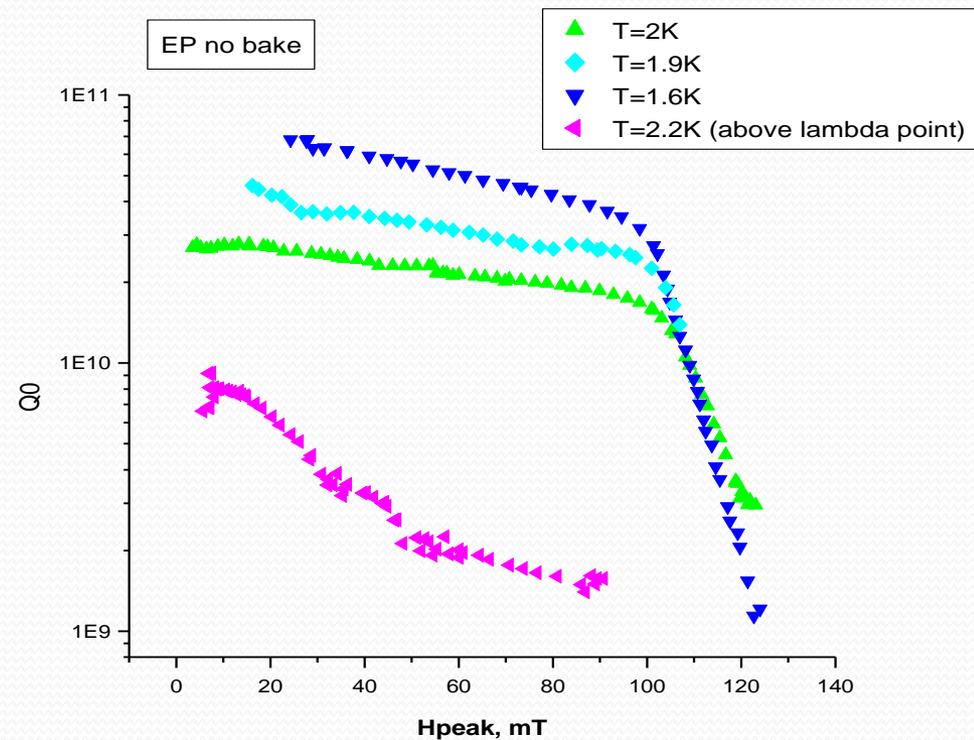
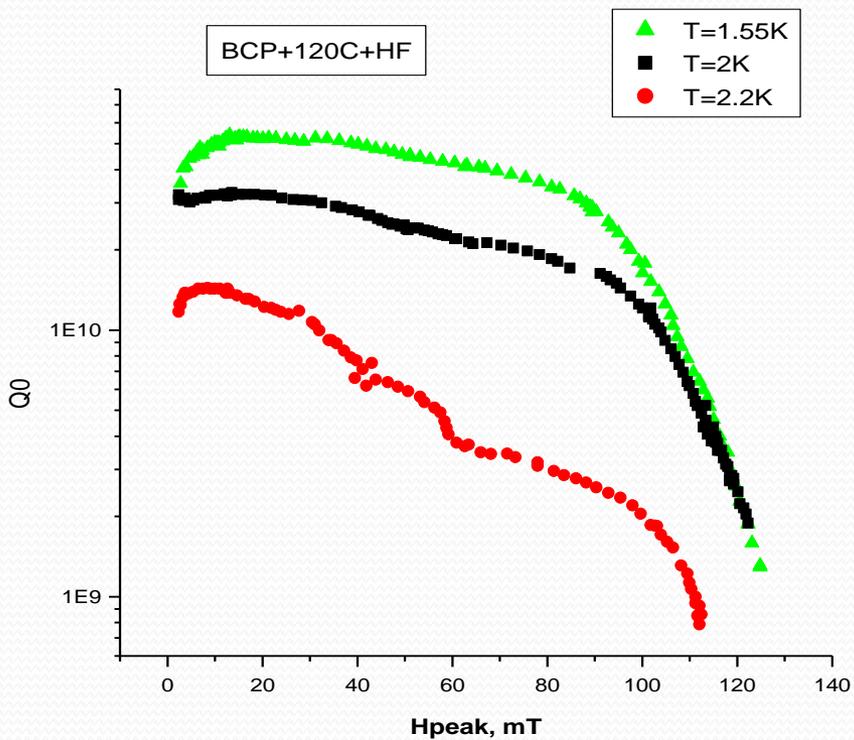
Some $Q_0(B_p)$ data plots



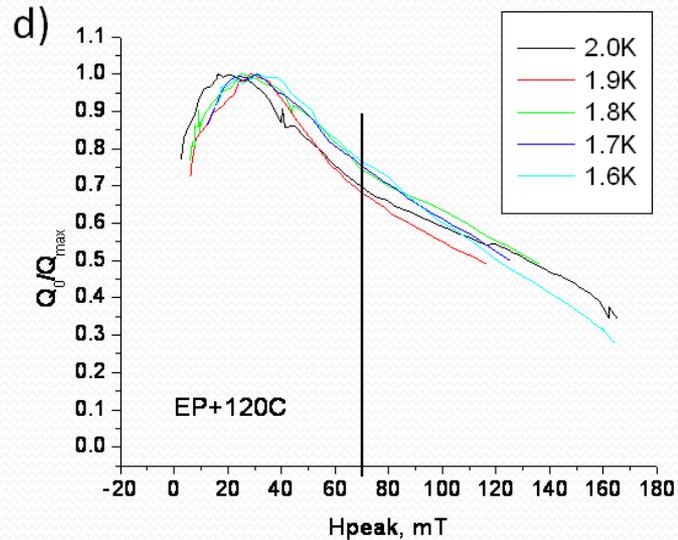
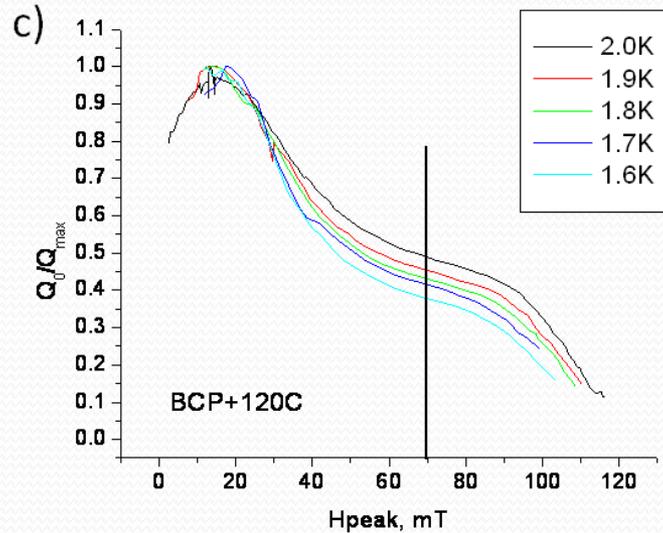
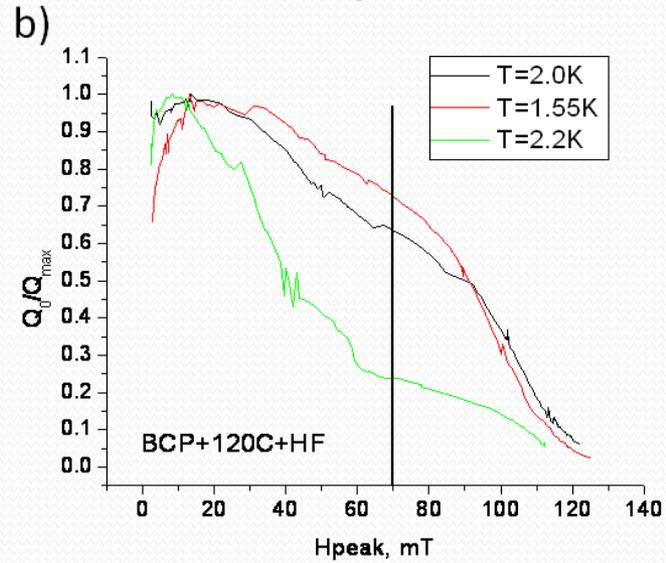
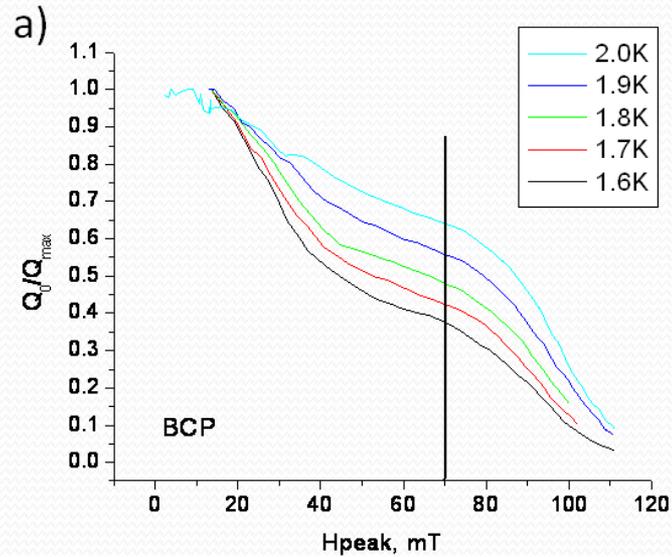
Recent effort at FNAL

- Bath temperature (1.6-2.2 K) and available surface treatments (BCP, EP, tumbling) effect on the medium field Q before/after mild baking [PAC'11 – Romanenko et al, TuPo85; SRF'11 – Romanenko, Ozelis, Wu]
- Tumbling improvement of Q_o in some cavities (C. Cooper)
- “Depth profiling of losses” via HF rinsing
 - Improvement in the low and medium field range – spin-off

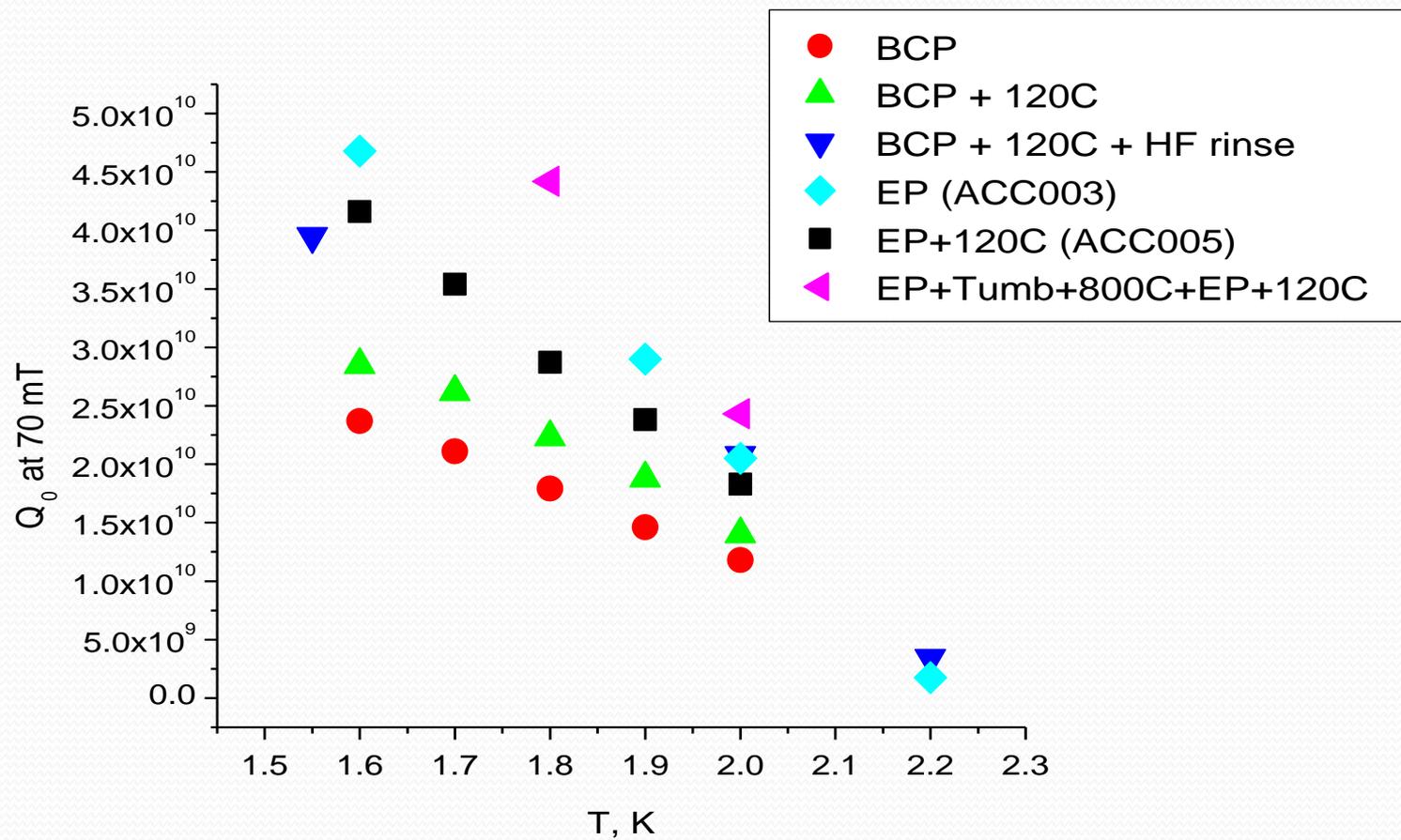
Confirmed $T > 2.17\text{K}$ not practical



Medium field Q-slope variation with treatments/bath temp

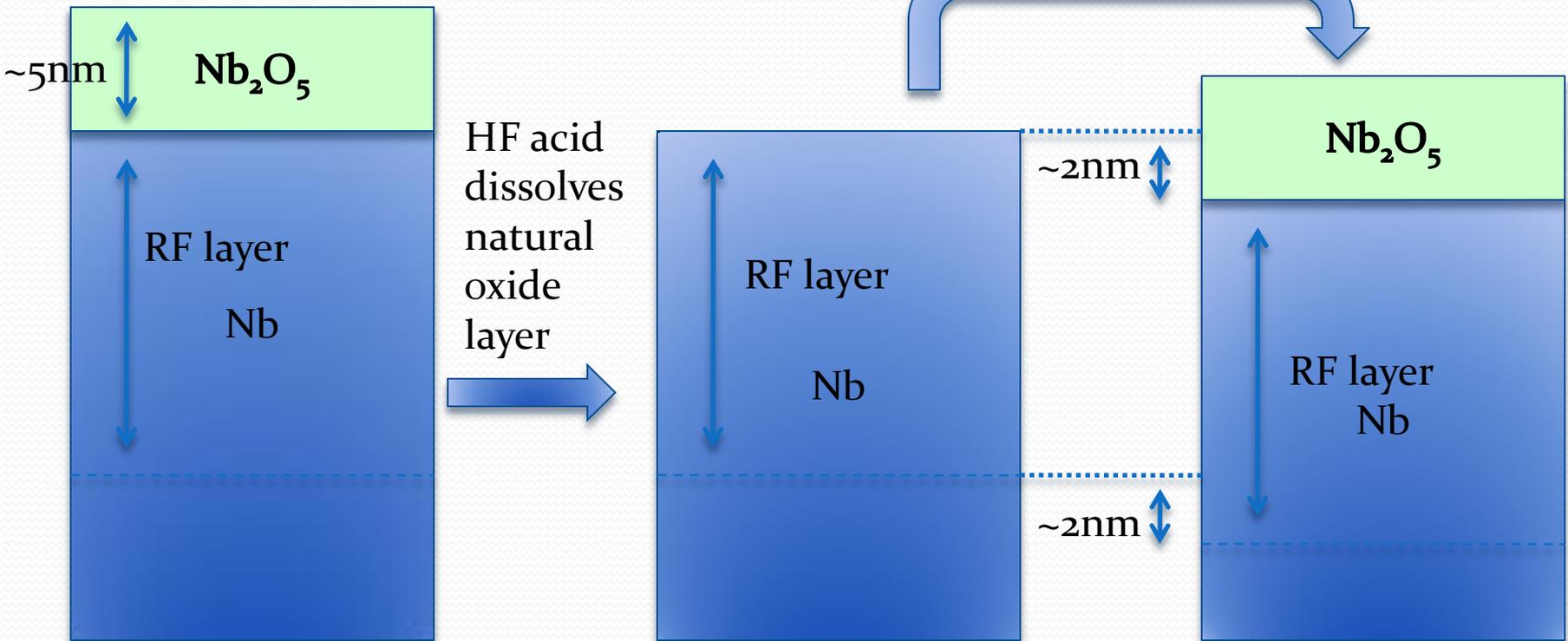


Q_0 at 70 mT



“Depth profiling of losses”

Water rinse grows
new oxide layer

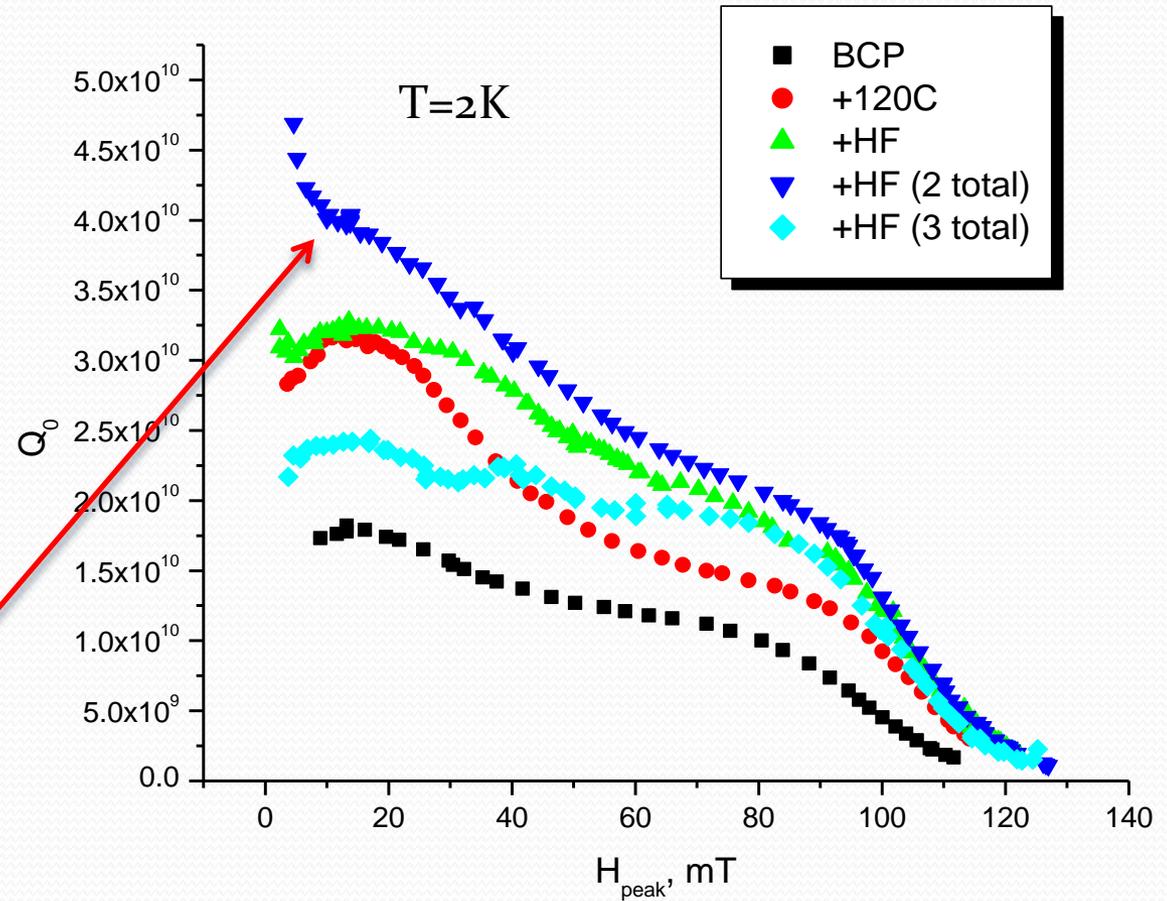


Each HF/water rinse step consumes about 2 nm of niobium from the top of the RF layer determining the surface resistance and moves deeper into the bulk – depth profiling of the losses

RF layer profiling by HF acid rinsing

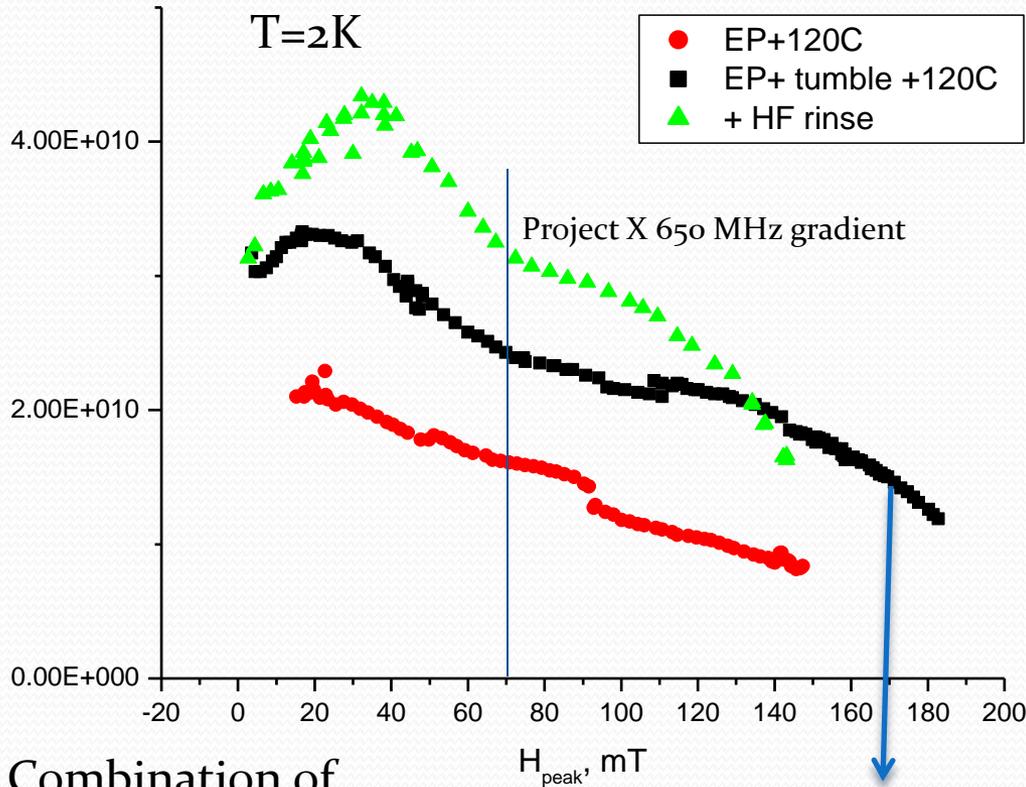
The goal of experiment:
what layer(s) determine
what regions of $Q(H)$
curve?

- Anodizing experiments indicated about 20 nm of mild baking modified layer (Eremeev et al, SRF'2005, TuAO8; Ciovati et al, PRST AB 10, 062002 (2007)) – ~expect 10 HF rinses to get everything back to original curve
- Highest Q_0 at low and medium fields after 2 rinses



Preliminary – first ~4 nm underneath oxide most important for medium field Q

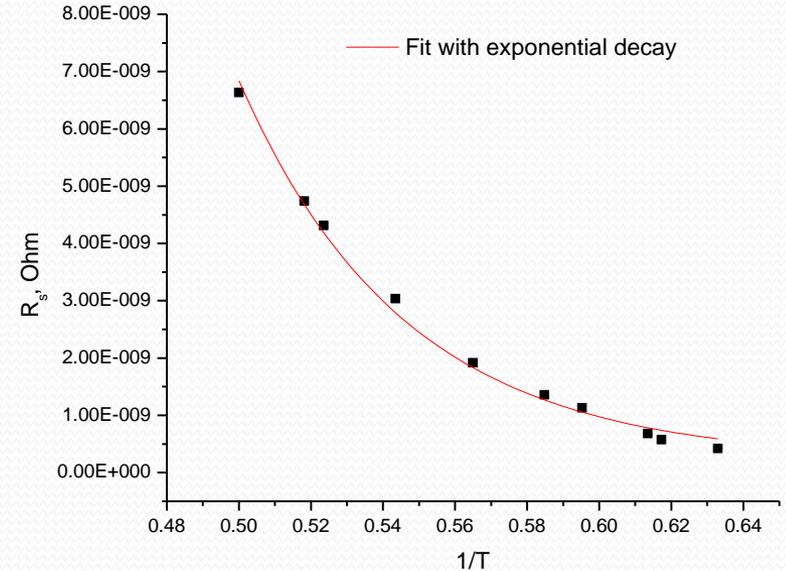
HF rinsing improvement on tumbled cavity



Combination of
smoothing and
removal of ~4nm
of Nb underneath
oxide

Best tumbled
single cell
(C. Cooper
research -
mirror finish)

Surface resistance estimate
for EP+tumble+800C+light
EP+120C+HF



R^2 0.99488

Parameter	Value	Error
y_0	2.25492E-10	1.81174E-10
A_1	3.56644E-4	3.34983E-4
t_1	0.04589	0.004

Residual resistance
<0.4 nOhm

Where do our findings lead us?

- After standard treatments
 - “Select” the RF layer with minimal medium field RF losses by sequential HF rinsing
- Alternative heat treatments
 - Explore temperature/duration range
- Correlate findings with samples investigation

Initial research plan (1)

- Experiments to be performed on
 - 1.3 GHz single cells
 - With temperature mapping after the system construction is completed
 - Later- single spoke resonators (325 MHz)
 - Different baseline treatment, even bigger possible margin of improvement
- Find the optimal number of HF rinsing steps for existing BCP, EP, tumbled cavities to maximize Q_o at fields of interest
~ 30 processes/RF tests
 - Confirm findings with different “nanoremoval” – anodizing ~ 10 processes/tests
 - Repeat the same with the large grain cavity ~ 10 processes/tests

Initial research plan (2)

- Explore heat treatments to modify first 40 nm (eliminate hydrogen, other interstitials and lattice defects) ~ 30 treatments/processes/tests
 - With/without passivation to avoid recontamination
 - With/without chemistry afterwards
 - Different temperatures: 100-1000C
 - Different durations
- “Witness” samples processed with cavities and analyzed using appropriate surface analytical tools
 - TEM – nanoscale imaging of the near-surface (~100 nm) structure
 - ERDA – nanoscale near-surface elemental distribution (especially H)
- **TOTAL = 80 processes/tests**

Resources needed

- Manpower
 - Scientist(s)/postdoc(s)/student(s) with SRF expertise
 - 1 FTE – HF rinsing/anodizing experiments
 - 1 FTE – Heat treatment experiments
 - 1 FTE – surface studies on samples following the same treatments
- Facilities time
 - ANL/FNAL processing facility
 - FNAL furnace
 - Vertical test stand (VTS)

Project constraints

- Competing for the same ANL/FNAL processing time
 - 9-cell 1.3 GHz CM production cavities – main project
 - 1-cell 1.3 GHz – ~1 cavity/week on average
 - ILC-related high field quench
 - High field Q-slope R&D
 - Tumbling
 - Vendor qualification
 - ARRA-funded initiatives (Cabot, Faraday)
 - Laser remelting
- ~ 80 tests/1 (test/week)=80 weeks (1.5 years) total time with 100% use of all available time for single cells
 - Subtract time for all other single-cell projects => ~10 years for the proposed project
- Manpower
 - 3 FTE total required

Conclusions

- Qo at medium fields -> direct impact on Project X costs (cryogenic plant)
- Feasible research program to investigate medium field Qo improvement
 - Manpower and facility time needed

Extended research plans

- Explore atomic layer deposition (ALD) or chemical vapor deposition (CVD) of some (yet to be determined) passivation layer – i.e. Al_2O_3
 - Collaboration with ANL or build on-site